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The effect of actual porous microstructure on the formation of dynamic necks and adiabatic shear bands

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Acknowledgements

This research has received funding from the European Union's Horizon 2020 research and innovation programme under the ERCEA grant agreement Nº 758056





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INVESTIGATION OF THE FRAGMENTATION BEHAVIOUR OF METALLIC MATERIALS:

THE ROLE OF GEOMETRIC AND MATERIAL DEFECTS IN PRINTED METALS



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SUMMARY

- **1. Introduction**
- 2. Objective and methodology
- **3. Porosity distribution**
- 4. Dynamic necking: formability
- **5. Dynamic shear banding: torsion**
- **6.** Conclusions

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1. Introduction RESEARCH MOTIVATION: PROTECTION APPLICATIONS Dynamic failure! (Undesired) Fragmentation **Increase the energy absorption** capacity of protective structures!! FIGURE 27.-Fragments recovered from a German 75 mm, high explosive shell.

RESEARCH MOTIVATION: METAL WORKING OPERATIONS

Dynamic failure!

(Undesired) Fragmentation



Sequence of underwater explosive forming operations (https://nptel.ac.in/courses/112107144/Metal%20Forming%20&%20 Powder%20metallurgy/lecture9/lecture9.htm)



Contact technique of explosive forming (https://nptel.ac.in/courses/112107144/Metal%20Formin g%20&%20Powder%20metallurgy/lecture9/lecture9.htm)







APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Rapid radial expansion of rings

Zhang and Ravi-Chandar (2006) University of Texas at Austin

Multiple necking and fragmentation



Some necks lead to fracture

Some necks are arrested

APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials



 $V_0 = 352 \text{ m/s} - 39 \le N \le 43$

Rapid radial expansion of rings

Zhang and Ravi-Chandar (2006) University of Texas at Austin

Multiple necking and fragmentation

Zhang and Ravi-Chandar 2006





Some necks lead to fracture

APPLICATIONS: laboratory experiment



APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Rapid radial expansion of rings

Nieto-Fuentes et al. (2022) University Carlos III of Madrid

Multiple necking and fragmentation







APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Rapid radial expansion of tubes

PhD Thesis Mathieu Xabier (2019) University of Lorraine

Multiple necking and fragmentation



FIGURE 1.13 – Images réalisées lors de l'expansion par voie détonique du cylindre en acier de l'expérience d'Olive et al. (1979) et exploitées par Jouve (2010). L'image 2 montre une déformation homogène. À $t = 25 \ \mu s$ (image 3), les premières localisations suivent les génératrices du cylindre. À $t = 34 \ \mu s$ (image 4), les strictions sont pleinement développées. Le nombre de strictions par mètre est évalué entre 125 et 150 par mètre par Jouve (2010). Les premières ruptures sont apparentes à $t = 52 \ \mu s$ (image 6).

APPLICATIONS: laboratory experiment

Increasing loading time

Dynamic failure!

Fragmentation

Ductile metallic materials

Rapid radial expansion of hemispherical shells

Mercier et al. (2010) University of Lorraine

Multiple necking and fragmentation







APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Axial penetration of thin-walled tubes

Nieto-Fuentes et al. (2022) University Carlos III of Madrid

Multiple necking and fragmentation











APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Axial penetration of thin-walled tubes

Nieto-Fuentes et al. (2022) University Carlos III of Madrid

Multiple necking and fragmentation

Crack path: The influence of voids



APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Axial penetration of thin-walled tubes

Nieto-Fuentes et al. (2022) University Carlos III of Madrid

> Multiple necking and fragmentation





1. Introduction APPLICATIONS: laboratory experiment **Dynamic failure! Specimen** after testing fracture **Ductile metallic materials** shear band scribe line unfractured) "Forced" shear band **Dynamic torsion of** thin-walled tubes gage length 2.5 mm

Marchand and Duffy (1988)



Stages in the formation of a shear band



APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Radial collapse of thick-walled tubes

Lovinger et al. (2015) RAFAEL

Multiple shear banding and fragmentation



Multiple shear bands develop from the inner radius of the cylinder



APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials

Radial collapse of thick-walled tubes



Lovinger et al. (2022) RAFAEL

Multiple shear banding and fragmentation



Multiple shear bands develop from the inner radius of the cylinder

APPLICATIONS: laboratory experiment

Dynamic failure!

Fragmentation

Ductile metallic materials





(c)

Radial collapse of thick-walled tubes

Experimental evidence of voids in shear bands



Xue et al. (2002)

Lovinger et al. (2015)









APPLICATIONS

Dynamic failure!

Fragmentation

APPLICATION TO ADDITIVE MANUFACTURED MATERIALS

Porosity distribution in AlSi10Mg cylindrical specimens

What is the effect on the dynamic energy absorption capacity of the material?



APPLICATIONS

Dynamic failure!

Fragmentation

NOVELTY

Many papers have investigated the influence of porosity on dynamic localization and fragmentation but ... ALL OF THEM USING HOMOGENIZED GURSON-TYPE CONSTITUTIVE MODELS TO DESCRIBE THE MATERIAL BEHAVIOUR

SPATIAL AND SIZE DISTRIBUTION OF ACTUAL VOIDS





1. Introduction MODELLING POROUS METALS Dynamic failure! Fragmentation **Modelling metallic porous metals** Homogenization: Gurson (1977) **Homogenization!!!** — No information on voids size distribution $\Phi\left(\Sigma_{h}, \Sigma_{e}, \bar{\sigma}, f\right) = \left(\frac{\Sigma_{e}}{\bar{\sigma}}\right)^{2} + 2q_{1}f\cosh\left(\frac{3q_{2}\Sigma_{h}}{2\bar{\sigma}}\right) - 1 - \left(q_{1}f\right)^{2}$ Porosity: single internal state variable

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2. Objective and methodology

Objective

Investigate the effect of ACTUAL porosity (material defects) on the formation of necks and shear bands: application to printed metals

Many papers have investigated using finite elements the role of defects in localization and fragmentation.... None included actual defects!!

The prevailing approach is to include idealized defects decreasing the yield stress of a number of elements, defining random variations of porosity..... Computationally cheap approach which IS NOT representative of actual defects!

2. Objective and methodology

Methodology

Finite element simulations with **actual** distributions of porosity

Own experiments are in process ...
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X-ray Computed Tomography

12 porous microstructures **AlSi10Mg**

Cylindrical specimens with diameter and height of 6 mm were **printed by Selective Laser Melting (SLM)**

Aluminium alloy **AlSi10Mg**, stainless steel **316L**, titanium alloy **Ti6Al4V** and Inconel **718L**











X-ray Computed Tomography

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X-ray Computed Tomography

12 porous microstructures

Log-Normal statistical distribution

	Al3Z	Al3XY	Al0.5Z	Al0.5XY	SS5Z	SS5XY	SS0.5Z	SS0.5XY	Ti0.5Z	Ti0.5XY	INC1Z	INC1XY
VF(%)	1.15	2.17	0.13	0.14	0.0290	0.0025	0.0007	0.0026	0.0033	0.0013	0.1363	0.0203
No. of voids	2500	5985	962	810	18	17	4	38	18	5	147	390
$d_{max} \; (\mu \mathrm{m})$	107	110.53	58.5	41.24	100	41.4	25.9	29.4	31.44	26.93	78.93	45.52
$d_{min} \; (\mu { m m})$.	8.02	8.00	8.1	8.01	7.44	7.40	8.06	8.01	7.44	8.27	7.45	6.36
$\mu~(\mu { m m})$	11.31	15.98	10.50	15.54	18.45	11.21	11.23	6.83	14.30	17.30	16.58	6.41
$dev \; (\mu { m m})$	5.03	4.57	3.81	4.37	9.85	5.13	5.04	6.70	9.31	5.03	7.71	4.56

Aluminium alloy AlSi10Mg, stainless steel 316L, titanium alloy Ti6Al4V and Inconel 718L



i M dea **3.** Porosity distribution materiales X-ray Computed Tomography SLM Φ6-L6 mm 3D reconstruction i M dea Material ImageJ materials (Schneider et al., X-ray 2012) Tomographs Computed Tomography Image (XCT) processing **Diameter of spheres** obtained from XCT Position of spheres obtained Voids assumed from Force Biased Algorithm as spheres (Bargiel and Moscinski, 1991) Marvi-Mashhadi et al. (2021) **3D model in Abaqus**

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Dynamic biaxial stretching of plates



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Dynamic biaxial stretching of plates







Mechanical behavior of the material

Linear elasticity and isotropic von Mises plasticity

Yield stress evolution $\sigma_Y = \sigma_0 + \sigma_K \left(\bar{\varepsilon}^p\right)^n \left(\frac{\dot{\bar{\varepsilon}}^p}{\dot{\varepsilon}_{ref}}\right)^m \left(\frac{T}{T_{ref}}\right)^{-\mu}$

Adiabatic conditions of deformation

Symbol	Property and units	Value
$ ho_0$	Initial density (kg/m^3)	7740
C_p	Specific heat $(J/kg K)$	460
E	Young modulus (GPa)	200
u	Poisson's ratio	0.3
σ_0	Initial yield stress (MPa)	175.67
σ_K	Hardening modulus (MPa)	530.13
n	Strain hardening exponent	0.167
m	Strain rate sensitivity exponent	0.0118
$\dot{arepsilon}_{ref}$	Reference strain rate (s^{-1})	0.01
μ^{2}	Temperature sensitivity exponent	0.51
T_{ref}	Reference temperature (K^{-1})	300
β	Taylor-Quinney coefficient	0.9

WE DO NOT CONSIDER THE SPECIFIC MECHANICAL **BEHAVIOUR OF THE** PRINTED METALS WHICH ARE ONLY USED TO OBTAIN POROUS MICROSTRUCTURES

 $\longrightarrow \dot{T} = \beta \frac{\bar{\sigma} \dot{\bar{\varepsilon}}^p}{\rho C_p}$



Dynamic biaxial stretching of plates

COMPARISON BETWEEN ACTUAL POROSITY AND HOMOGENIZED POROSITY



Dynamic biaxial stretching of plates

COMPARISON BETWEEN ACTUAL POROSITY AND HOMOGENIZED POROSITY

Evolution of the necking strain with microstructural features



The same general trends for other loading paths and strain rates

Increasing initial void volume fraction and maximum void diameter lead to a decrease of the necking formability



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Dynamic biaxial stretching of plates

The homogenized porosity calculations provide results for the necking strain which are in qualitative agreement with the actual porosity simulations, the quantitative differences being generally less that 25%

Initial void volume fraction and **maximum voids size** affect the necking strain, with the initial void volume fraction playing a critical role in the specimen ductility

The **number** and the **location** of necks predicted by the homogenized porosity approach and the actual porosity calculations are <mark>very similar</mark> for all loading rates and loading paths investigated

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5. Dynamic shear banding: torsion Dynamic twisting of thin-walled tube Representative Volume Element: strip subjected to angular velocity Fine mesh to Finite element model – Microstructure: INC1Z-R1 capture the shape of Voids distribution Mesh the voids **Top surface:** Application of periodic force-displacement boundary condition in measurement opposed sides Large **Cut-view** computational cost One over thirty-two of the gauge section Lateral edge: measurement zoom Fine mesh path for the shear band around the void evolution Random spatial distribution of void sizes (a) (b)

Mechanical behavior of the material

Linear elasticity and isotropic von Mises plasticity

Yield stress evolution $\sigma_Y = \sigma_0 + \sigma_K \left(\bar{\varepsilon}^p\right)^n \left(\frac{\dot{\bar{\varepsilon}}^p}{\dot{\varepsilon}_{ref}}\right)^m \left(\frac{T}{T_{ref}}\right)^{-\mu}$

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5. Dynamic shear banding: torsion

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Onset and development of shear band



5. Dynamic shear banding: torsion

Comparison with fully dense material

Porosity leading to early shear localization: loss of load carrying capacity

Porous microstructure breaks the symmetry of the problem: shear band incepted by the greatest void of the microstructure





Normalized axial coordinate, Z











5. Dynamic shear banding: torsion The effect of porous microstructure Additional insights: THE MEAN VOIDS SIZE The void volume fraction is constant **INCREASING THE VOIDS** 250 SIZE LEADS TO EARLY LOSS OF LOAD CARRYING **CAPACITY** 200 f_=0.13 % Shear force, F (N) μ=10 μm 150 μ=30 μm 100 50 Material: Titanium $\dot{\gamma}_0 = 1000 \text{ s}^{-1}$ Parent microstructural realization: INC1Z-R1 0 0.02 0.04 0.06 0.1 0.08 0 67 Angular displacement, ϕ (radians)

5. Dynamic shear banding: torsion

Dynamic twisting of thin-walled tube

The spatial distribution of voids affects the morphology of the shear localization pattern, as the main shear band is generally nucleated at the location of the largest pore of the microstructure

The size distribution of voids affects the morphology of the shear localization pattern such that when the pores show large differences in size, the main shear band stands out over the secondary localizations

The **maximum voids size** is the main feature of the porous microstructure affecting the specimen ductility, over and above the void volume fraction

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6. Conclusions

Main outcomes

Lay the groundwork of a microstructurally-informed finite element strategy to describe dynamic localization and fracture of porous materials

6. Conclusions

Some limitations

We still lack of a COMPRENHENSIVE EXPERIMENTAL VALIDATION within a wide range of strain rates and material behaviours

The computational time is LARGE!
6. Conclusions

Related publications

Vishnu A. R., Marvi-Mashhadi M., Nieto-Fuentes J. C., Rodríguez-Martínez J. A. *New insights into the role of porous microstructure on dynamic shear localization*. International Journal of Plasticity. 2022; 148: 103150

Marvi-Mashhadi M., Vaz-Romero A., Sket F., Rodríguez-Martínez J. A. *Finite element analysis to determine the role of porosity in dynamic localization and fragmentation: Application to porous microstructures obtained from additively manufactured materials.* **International Journal of Plasticity.** 2021, 143, 102999





6. Conclusions

Related publications

Vishnu A. R., Nieto-Fuentes J. C., Rodríguez-Martínez J. A. *Shear band formation in porous thin-walled tubes subjected to dynamic torsion*. Submitted for publication.

Nieto-Fuentes J. C., Jacques N., Marvi-Mashhadi M., N'souglo K. E., Rodríguez-Martínez J. A. *Modeling dynamic formability of porous ductile sheets subjected to biaxial stretching: Actual porosity versus homogenized porosity*. Submitted for publication.





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Dynamic compression of thick-walled tube

Representative Volume Element: external pressure loading



4. Dynamic necking: formability

Mechanical behavior of the material

Linear elasticity and isotropic von Mises plasticity

Yield stress evolution $\sigma_Y = \sigma_0 + \sigma_K \left(\bar{\varepsilon}^p\right)^n \left(\frac{\dot{\bar{\varepsilon}}^p}{\dot{\varepsilon}_{ref}}\right)^m \left(\frac{T}{T_{ref}}\right)^{-\mu}$

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 $\longrightarrow \dot{T} = \beta \frac{\bar{\sigma} \bar{\varepsilon}^p}{\rho C_p}$

Onset and development of MULTIPLE shear bands

Evolution of the porous microstructure



Onset and development of MULTIPLE shear bands

Comparison with fully dense material



+1.500e+00

375e+00 50e+00

25e+00

00e+00 50e-01

50e-01 000e-01

750e-01 2.500e-01

250e-01

+0.000e+00

The effect of thermal softening

The porous microstructure **ALONE** does not trigger the localization (the same for the torsion of thin-walled tubes)









Dynamic compression of thick-walled tube

Microstructural porosity favors the formation and development of shear bands in temperature softening materials

For temperature independent materials, microstructural porosity alone does not trigger the formation of shear bands

Increasing the maximum diameter of the voids favors early shear localization and decreases the number of shear bands

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- 6. Dynamic shear banding: thick-walled cylinder collapse
- 7. Conclusions

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8. Plastic shock waves



8. Plastic shock waves



8. Plastic shock waves



Ring expansion finite element model with actual distribution of porosity

Inner and outer radii $R_{in} = 15 mm$ and $R_{out} = 15.5 mm$ Constant radial velocity V_r varying between 50 and 500 m/s



THE INFLUENCE OF MICROSTRUCTURAL REALIZATION IN THE NECKING AND FRAGMENTATION PROCESSES



THE INFLUENCE OF LOADING VELOCITY IN THE NECKING AND FRAGMENTATION PROCESSES



INCREASING VELOCITY INCREASES THE NUMBER OF NECKS AND FRAGMENTS

Deterministic approach: inertia promotes the development of necking wavelengths which are smaller and more dominant as the strain rate increases

THE INFLUENCE OF LOADING VELOCITY IN THE NECKING AND FRAGMENTATION PROCESSES



THE INFLUENCE OF LOADING VELOCITY IN THE NECKING AND FRAGMENTATION PROCESSES

